# The Properties of Degradation in Epoxy Composites according to Electrical Stress

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#### Abstract

The electrical degradation phenomena of epoxy composites to be used as a molding material for transformers were studied. The electrets were first manufactured by applying high voltages to five kinds of specimens given a mixing rate, and then TSC(Thermally Stimulated Current) values at the temperature range of  $-160 \sim 200$  [°C] were measured from a series of experiments. The behaviour of carrier and its origin in epoxy composites were examined, respectively. And various effects of electrical degradation on epoxy composites were also discussed in this study.

Key Words: degradation phenomena, Epoxy Composites, TSC(Thermally Stimulated Current) .

### 1. INTRODUCTION

As the scale of a power delivery system is increased, the insulating design techniques such as a materialization of high electric field and a security of reliability in insulating composition are required accordingly. Previous works have been performed to develop new composites whose insulating properties are better than existing materials and to examine the impact of the degradation phenomena on electrical properties in epoxy composites which are generally employed for molding materials in transformers.

The main purpose of this research is to study the TSC values in electrical stressed epoxy composites and interrelationship between electrical stress and material degradation. The TSC values associated with the determination of the currents which were released from poled specimens were measured as a function of temperature.

Poling was accomplished by placing the specimen under a high DC stress and followed by cooling with liquid nitrogen to freeze the molecular motion of the polymer, resulting in immobilizing the charges. When the specimens are warmed slowly, the trapped charges are released. By measuring the amount of charge released and their release characteris –tic temperature, inferences on the space charges which form in the polymer can be made.

It has been noticed that peak temperatures for a given heating rate depend on a variety of factors such as age of the specimen, thermal history, and charge density. To investigate the effect of the electrical degradation on properties of epoxy composites, the TSC spectra were measured as a function of temperature and the various aspects of electrical degradation effects were analyzed.

## 2. EXPERIMENTAL

## 2-1. Specimens

The specimens used in this study were

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designed to have a constant mixing rate in the epoxy resin of Bisphenol-A type and the McTHPA(Methyl Tetra Hydro Phthalic Anhydride). Bisphenol-A type is in liquid phase at room temperature, and the McTHPA is a hardner of acid hydro-anhydride system. Also, The DY-040 and the  $SiO_2$  with amount of  $SiO_2$  with amount of  $SiO_3$  are then added to improve both the impact strength and the machinery strength, and the silane coupling agents, which belong to an amino silane system (chemical name;  $N-(N-(\beta-Aminoetyl)-Aminopropyl-trimetoxy-Silane)$ ) have been used for surface treatment of fillers.

## 2-2. Design of Mixing Ratio

Since the electrical and physical properties of epoxy composites depend greatly on the mixing ratio among resin, hardner and filler, as well as the hardening conditions [1]-[3], these aspects need to be considered in manufacturing specimens. The design parameters and their condition used for this study are summarized in Table 1.

Table 1. Mixing Ratio [wt%]

Samples	Ероху	Hardener	DY -040	Filler	Curing Condition
H 70FN	100	70	10	0	▶ 1st Curing; 100[℃] × 4[hr] ▶ 2nd Curing; 140[℃] × 10[hr] ▶ Deg. Curing; 140[℃] × 240[h
H100FN	100	100	10	0	
H130FN	100	130	10	0	
H100F60	100	100	10	60	
SH100F60	100	100	10	60	

## 2-3. Experimental Method

After the samples were degraded by appling high voltages, 22.9[kV/cm], during 5[hr], the electrets<sup>[4]~[5]</sup> were formed by appling a electric field, 10[kV/cm], to epoxy composites and then TSC spectra were measured at the temperature range of -160~200[°C] with a step of 5[°C/min]. A Schematic drawing of the experimental apparatus to be used for TSC values is given in

Fig. 1.

The specimens used in this study were designed to have a constant mixing rate in the epoxy resin of Bisphenol-A type and the MeTHPA(Methyl Tetra Hydro Phthalic Anhydride).

Bisphenol-A type is in liquid phase at room temperature, and the MeTHPA is a hardener of acid hydro-anhydride system. Also, the DY-040 and the SiO<sub>2</sub> with amount of 5[wt%] are added to improve both the impact strength and the machinery strength. and the silane coupling agents, which belong to an amino silane system (chemical name; N-(N-( $\beta$ -Aminoethyl)-Aminopropyl-trimetoxy-Silane)), have been used for surface treatment of fillers<sup>[3],[4]</sup>.

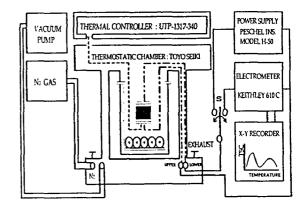


Fig. 1. Experimental Apparatus

## 3. RESULTS AND DISCUSSION

# 3-1. TSC Spectra due to Forming Electric Field

Fig. 2 shows TSC spectra formed at 5, 10, and 30[kV/cm], which were observed from electrets for the 1st cured( $100[C]\times4[hr]$ ) H100FN specimens. Similarly, Fig. 3 includes TSC spectra formed at 5, 10, and 30[kV/cm], using the 2nd cured( $140[C]\times1[hr]$ ) H100FN specimens. From the experimental study, the five peaks of  $\delta$ ,  $\gamma$ ,  $\beta$ ,  $\alpha_1$  and  $\alpha_2$  were obtained at the temperature of -100[C], -40[C], 20[C], 100[C] and 130[C], respectively. The first three peaks which are observed at the region of low

temperature below the glass transition temperature  $(T_R)$  are relaxation peaks due to the action of side chain or terminal radical. Also, both the  $\alpha_1$  peak near  $100[\,^{\circ}\!\!\!C]$  and the  $\alpha_2$  peak above  $T_R$  are peaks caused by ionic space charge. The causes of these peaks are as follows:

- (1) Since the  $\delta$  peaks have been suggested by either a methyl radical(-CH<sub>3</sub>-) or a non-acted epoxy radical, the overall size of peaks is not so big. From the molecular structure, it is observed that the methyl radical is independently operated in proportion to the level of hardener added and it also affects curing reaction.
- (2) The  $\gamma$  peak is produced by the mixture of hydroxyether radical, methoxy radical(CH-OH), aliphatic ether radical(-CH-O-), and aromatic ether radical(-COO-CH-) which is related to hardner. Also, as the amount of hardner is increased, the size of peak is getting bigger likewise the  $\delta$  peak.
- (3) The  $\beta$  peak appears to be resulted from aromatic ether radical (-COO-CH-) which is thermally oxided through the procedure of maximum reaction.
- (4) The  $\alpha_1$  peak is believed to be the effects from the virtue of molecule which is related to glass transition temperature.

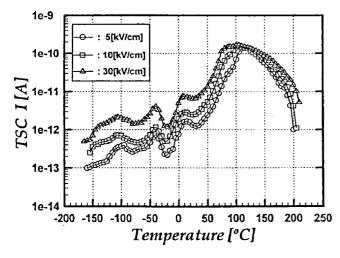


Fig. 2. The Dependence on Forming Electrical Strength in the 1st Cured H100FN Samples

(5) The  $\alpha_2$  peak which is observed at the highest temperature is generated by the ionic space charge which is accumulated by impurities or inside defects.

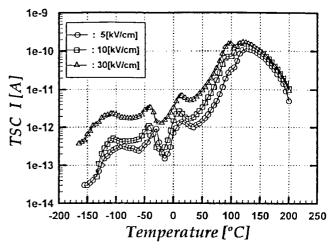


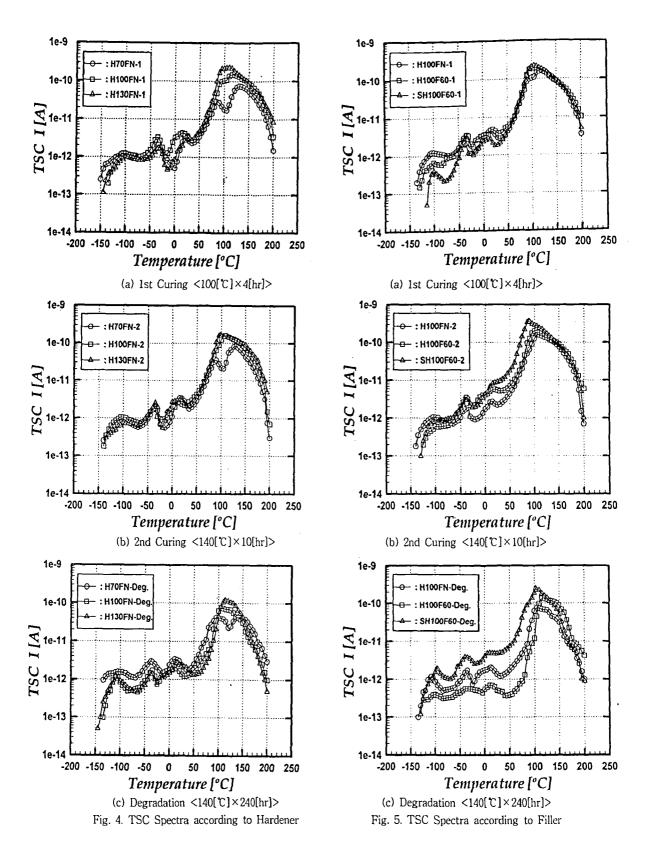
Fig. 3. The Dependence on Forming Electrical Strength in the 2nd Cured H100FN Samples

### 3-2. TSC Spectra dued to Electrical Stress

TSC spectra, which are divided according to curing time and observed from electrets degraded by appling electrical stress, are shown in Fig. 4 and 5.

It is noted that in the case of the longest curing time, the size of the  $\alpha$  peak which is shown above  $100[\,^\circ\!\!C\,]$  is smaller than those of 1st cured specimen and 2nd cured specimen. Particularly, the amplitude among samples is not significant below  $T_g$  and this result implies that the radicals shown at low temperature are affected by electrical stress. [8]-[10]

On the other hand, molecular motion become dull because of excessive hardener, and the peak size of H130FN is decreased above  $T_{\rm g}$ . Also,  $I_{\rm m}$  of SH100F60 is reduced above  $T_{\rm g}$  but that of H100F60 is not reduced because of improving deformation of interfacial state.



### 4. CONCLUSION

In order understand the degradation characteristics of epoxy composites due to electrical TSC spectra were measured temperature range of  $-160 \sim 200$  [°C]. From a series of experiments, five peaks including  $\delta$ ,  $\gamma$ ,  $\beta$ ,  $\alpha_1$  and  $\alpha_2$  were obtained at the temperature of  $-100^{\circ}$ C,  $-40^{\circ}$ C,  $20^{\circ}$ C,  $100^{\circ}$ C, and 130℃, respectively. Results of this study indicates that (1) the  $\delta$  peak is resulted from the methyl radical; the y peak is produced due to the mixture by hydroxyether radical, methoxy radical, aliphatic ether radical and aromatic ether radical; the  $\beta$  peak is obtained from aromatic ether radical; the  $\alpha_1$  peak is possibly due to molecule which is related to glass transition temperature; and the  $\alpha_2$  peak seems to turn up by the ionic space charge, and (2) in the case of the longest curing time, the size of the  $\alpha$  peak which is shown above 100[°C] is smaller than those of 1st cured specimen and 2nd cured specimen. Particularly, the amplitude among samples is not significant below  $T_g$  and this result implies that the radicals shown at low temperature are affected by electrical stress. And, molecular become dull because of excessive hardener, and the peak size of H130FN is decreased above Tg. Also, Im of SH100F60 is reduced above T<sub>g</sub> but that of H100F60 is not reduced because of improving deformation of interfacial state.

#### REFERENCES

- [1] G. M. Sessler, "Electret", Springer-Verlag, pp.81~201, 1980
- [2] J. P. T Fillard and J. Van Turnhout, "Thermally Stimulated Processes in Solids, New Prospects", Elsevier Scientific Pub., pp.139~180, 1977
- [3] T. Takahama, O. Hayashi, F. Sato, "Electric Strength of Epoxide Resins and its Relation to the Structure", J. Appl. Poly. Sci., Vol.26, pp.2211~2220, No.5, 1981
- [4] J. V. Duffy and G. F. Lee, "The Effect of Steric Hinderance on Physical Properties in an

- Amine-Cured Epoxy", J. Appl. Poly. Sci., Vol.35, pp.1367~1375, 1988
- [5] M. Ochi, M. H. Okasaki, M. Shimbo, "Mechanical Relaxation Mechanism of Epoxide Resins Cured Aliphatic Diamines", J. Poly. Sci., Phys. Ed., Vol.20, pp.689~699, 1982
- [6] O. Delatycki, J. C. Shaw and J. G. Williama, "Viscoelastic Properties of Epoxy-Diamine Networks", J. Poly. Sci., A2, Vol.7, pp.753~762, 1969
- [7] L. Simoni, "A General Approach to the Endurance of Electrical Insulation under Temperature and Voltages", Colloid & Polymer Sci., 260, pp.297~302, 1982
- [8] R. Schifani, "Surface Discharge Effects on Dielectric Properties of Epoxy Resins", IEEE Trans. Electr. Insul., Vol.EI-18, pp.504~512, No.5, Oct., 1983
- [9] S. S. Sastry, G. Satyanandam, "Effects of Fillers on Electrical Properties of Epoxy Composites", J. Appl. Poly. Sci., Vol.26, pp.1607 ~1615, 1988
- [10] J. J. O'Dwyer, "The Theory of Electrical Conduction and Breakdown in Solid Dielectrics", Clarendon Press. Oxford, 1973