

## High density plasma etching of novel dielectric thin films: Ta<sub>2</sub>O<sub>5</sub> and (Ba,Sr)TiO<sub>3</sub>

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**Abstract** Etch rates up to 120 nm/min for Ta<sub>2</sub>O<sub>5</sub> were achieved in both SF<sub>6</sub>/Ar and Cl<sub>2</sub>/Ar discharges. The effect of ultraviolet (UV) light illumination during ICP etching on Ta<sub>2</sub>O<sub>5</sub> etch rate in those plasma chemistries was examined and UV illumination was found to produce significant enhancements in Ta<sub>2</sub>O<sub>5</sub> etch rates most likely due to photoassisted desorption of the etch products. The effects of ion flux, ion energy, and plasma composition on (Ba,Sr)TiO<sub>3</sub> etch rate were examined and maximum etch rate ~90 nm/min was achieved in Cl<sub>2</sub>/Ar ICP discharges while CH<sub>4</sub>/H<sub>2</sub>/Ar chemistry produced extremely low etch rates (≤ 10 nm/min) under all conditions.

**Key words** High density plasma etching, Dielectric thin film, Ta<sub>2</sub>O<sub>5</sub>, (Ba,Sr)TiO<sub>3</sub>, ICP etching, UV illumination

### 1. Introduction

There has been extensive effort over the past decade to develop novel dielectric thin films which can be a replacement for SiO<sub>2</sub> and the leading candidates are Ta<sub>2</sub>O<sub>5</sub> and (Ba,Sr)TiO<sub>3</sub> based on their high dielectric constant (Ta<sub>2</sub>O<sub>5</sub> : 25~35, (Ba,Sr)TiO<sub>3</sub> : 200~500), low leakage current and high breakdown voltage characteristics [1-3]. These films have many application fields include a dielectric layer for storage capacitors in dynamic random access memory (DRAM) and gate insulators in metal oxide field effect transistors (MOS-FET) [4-5]. Recently, Ta<sub>2</sub>O<sub>5</sub> has found new application fields as an etch mask during surface or bulk micro-machining of Si, as an insulating layer in thin film electroluminescent display devices, and as a detection layer in biological and chemical sensors [6-8]. However, relatively little work has been done on dry etching of these thin films, which is preferred method of forming small structures due to the difficulty in wet etching while dry etching process for conventional SiO<sub>2</sub>-based dielectric structures are well-developed. Previous reports have found relatively low etch rates for Ta<sub>2</sub>O<sub>5</sub> in fluorocarbon-based plasma chemistries such as CF<sub>4</sub>, C<sub>2</sub>F<sub>6</sub>, CHF<sub>3</sub> and CF<sub>3</sub>Cl [9]. There have been two basic classes of electrode materials employed to date, namely, those based on elemental metals, predominantly Pt or those based on metallic

oxides such as IrO<sub>2</sub>, RuO<sub>2</sub>, and high T<sub>c</sub> superconductors [10]. The metallic oxides have a potential advantage in improving the fatigue performance of capacitors. In this work, LaNiO<sub>3</sub> was chosen as the metallic oxide for use with (Ba,Sr)TiO<sub>3</sub> films, since it displays several advantages as an electrode material [11-13].

In this paper we report on high density plasma etching of Ta<sub>2</sub>O<sub>5</sub> and (Ba,Sr)TiO<sub>3</sub> thin films in a variety of different plasma chemistries, namely SF<sub>6</sub>/Ar, SF<sub>6</sub>/O<sub>2</sub>, Cl<sub>2</sub>/Ar, and CH<sub>4</sub>/H<sub>2</sub>/Ar, and the significant enhancement in Ta<sub>2</sub>O<sub>5</sub> etch rate by ultraviolet (UV) light illumination during inductively coupled plasma (ICP) etching.

### 2. Experimental

Around 100 nm thick amorphous Ta<sub>2</sub>O<sub>5</sub> and 200~300 nm thick polycrystalline (Ba,Sr)TiO<sub>3</sub> films were deposited on (100) Si substrates by either plasma enhanced chemical vapor deposition (PECVD) at 350°C or pulsed laser deposition (PLD, KrF excimer laser, 5 Hz pulse frequency) at 650°C. The precursors used for Ta<sub>2</sub>O<sub>5</sub> were Ta(C<sub>2</sub>H<sub>5</sub>O<sub>5</sub>) and O<sub>2</sub> with total flow rates of 200 standard cubic centimeters per minute (sccm) and pressed powder targets were used for (Ba,Sr)TiO<sub>3</sub> deposition.

The samples were patterned with either Apiezon wax or photoresist and were etched in a Plasma-Therm ICP 790 reactor. The plasma is sustained in a three-turn, cylindrical geometry source operating at 2

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MHz and powers up to 1000 W. The samples were thermally bonded to a Si carrier wafer on He back-side cooled, rf powered (13.56 MHz) chuck, at powers up to 350 W. The process pressure was held at 2~5 mTorr and gas loading into the source was controlled through mass flow controllers at a typical load of 15~20 sccm. For UV illumination during ICP etching, a Hg arc lamp (400 W) was installed on 1 in. diameter quartz window on top of the ICP source, ~20 cm from the sample position, and sample heating due to the UV illumination is minimal (< 10°C). The etch depths were examined by TENCOR stylus profilometry measurements after the removal of the mask, and used to calculate the etch rates. The error in these measurements is approximately ±10 %. The surface morphology and surface roughness of selected samples were examined with Atomic Force Microscopy (AFM, Digital Instrument Nanoscope III) using a Si tip in tapping

mode. The selectivity of etch was calculated for Si over Ta<sub>2</sub>O<sub>5</sub>.

### 3. Results and Discussion

#### 3.1. ICP etching of Ta<sub>2</sub>O<sub>5</sub>

Figure 1 shows a comparison of Ta<sub>2</sub>O<sub>5</sub> and Si etch rates, with the resultant selectivities, as a function of discharge composition in SF<sub>6</sub>/Ar (left) and SF<sub>6</sub>/O<sub>2</sub> (right). The etch rates for Ta<sub>2</sub>O<sub>5</sub> increase more slowly with increasing SF<sub>6</sub> concentration than do the rates for Si, with the consequence that the maximum selectivity for Si over Ta<sub>2</sub>O<sub>5</sub> is achieved in pure SF<sub>6</sub> discharges. The dc self-bias increases in both chemistries as the SF<sub>6</sub> concentration increases, indicating that the positive ion density is decreasing. Note that Ta<sub>2</sub>O<sub>5</sub> etch

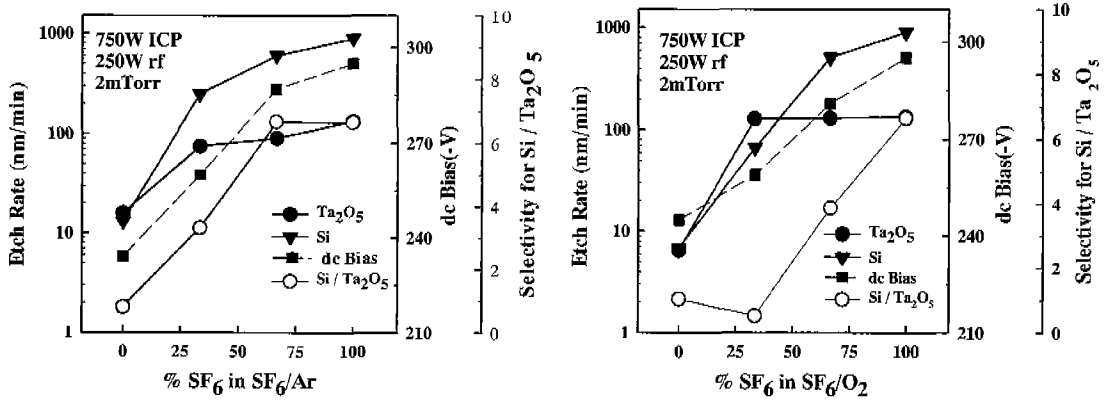


Fig. 1. Etch rates and selectivities for Si over Ta<sub>2</sub>O<sub>5</sub> in SF<sub>6</sub>/Ar (left) and SF<sub>6</sub>/O<sub>2</sub> (right) ICP discharges (750 W source power, 250 W rf chuck power, 2 mTorr), as a function of SF<sub>6</sub> concentration.

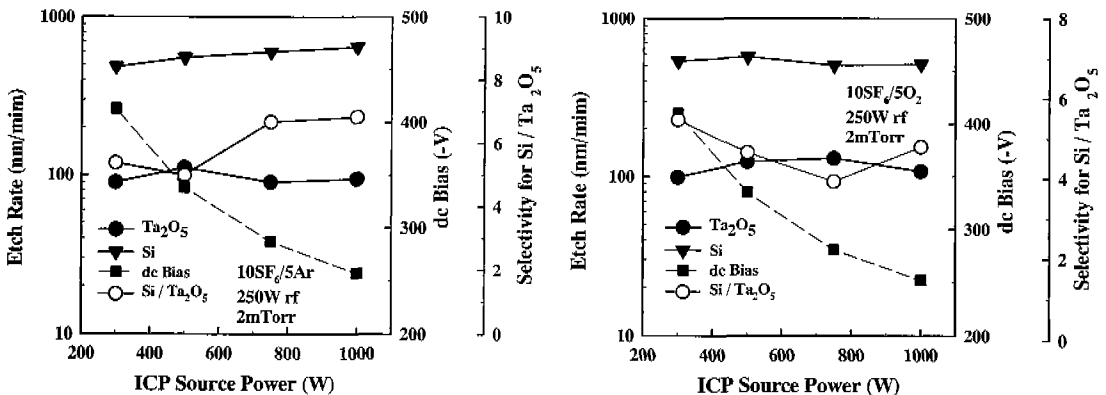


Fig. 2. Etch rates and selectivities for Si over Ta<sub>2</sub>O<sub>5</sub> in 10SF<sub>6</sub>/5Ar (left) and 10SF<sub>6</sub>/5O<sub>2</sub> (right) ICP discharges (250 W rf chuck power, 2 mTorr), as a function of ICP source power.

rates of ~20 nm/min are achieved in the SF<sub>6</sub>-based mixtures, at dc self-biases in the range of 215 to 290 V. We could not detect the etch products for Ta<sub>2</sub>O<sub>5</sub> with optical emission spectroscopy (OES), but assume they are probably TaF<sub>x</sub> and O<sub>2</sub>. In the case of Si, we readily observed the SiF<sub>x</sub> etch products, with emission lines in the range of 400~430 nm.

The effect of ICP source power on the Ta<sub>2</sub>O<sub>5</sub> etch rates is shown in Fig. 2 for fixed plasma composition. The Si etch rate increases either modestly or not all over the range of 300~1000 W, while the Ta<sub>2</sub>O<sub>5</sub> etch rate tends to decrease at the higher powers. This decrease is at least partially caused by the fall-off in dc self-bias, which is suppressed as the positive ion density in the discharge increases at higher powers. It is clear that because Si etches in atomic fluorine even without ion bombardment, whereas Ta<sub>2</sub>O<sub>5</sub> does not, there will always be a faster etch rate for Si in non-polymer-forming plasma chemistries.

In high density plasma etching the rf chuck power controls the incident ion energy approximately equal to the sum of the plasma potential (roughly 25 V in this tool, from Langmuir probe measurements) and the dc self-bias. It is expected an increase in rf chuck power increases the etch rates of both Ta<sub>2</sub>O<sub>5</sub> and Si because of the improved efficiency of ion-assisted reactions. Figure 3 shows the trend consistent with this expectation for both SF<sub>6</sub>/Ar and SF<sub>6</sub>/O<sub>2</sub>. Selectivity for Si over Ta<sub>2</sub>O<sub>5</sub> is found to decrease with increasing ion energy because of the larger contribution of physical process relative to chemical component of the etching.

The surface morphologies of the etched Ta<sub>2</sub>O<sub>5</sub> were examined by AFM and Fig. 4 shows the measured

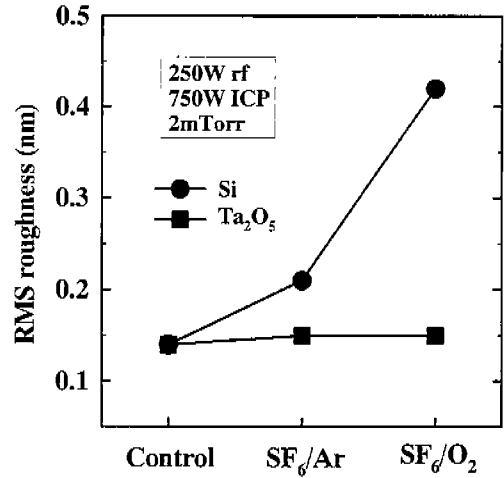


Fig. 4. RMS roughness values of Ta<sub>2</sub>O<sub>5</sub> and Si surfaces before and after etching in 10SF<sub>6</sub>/5Ar and 10SF<sub>6</sub>/5O<sub>2</sub> ICP discharges (750 W source power, 250 W rf chuck power, 2 mTorr).

root-mean-square (RMS) roughness measured over 1 mm<sup>2</sup> area after exposure to SF<sub>6</sub>/Ar or SF<sub>6</sub>/O<sub>2</sub> discharges. The Ta<sub>2</sub>O<sub>5</sub> surface roughness values are unchanged from the unetched control values, whereas the Si shows some etch-induced texture structure on the surfaces. In each case the etch was performed for 30 secs corresponding to etch depths of ~45 nm for Ta<sub>2</sub>O<sub>5</sub> and ~250 nm for Si.

Figure 5 shows the effects of source power (left) and rf chuck power (right) on the Ta<sub>2</sub>O<sub>5</sub> and Si etch rates at fixed plasma composition (10Cl<sub>2</sub>/5Ar). The etch rates for both materials decrease at high values of either parameter, a trend often observed in high density plasma etching and often ascribed to ion-

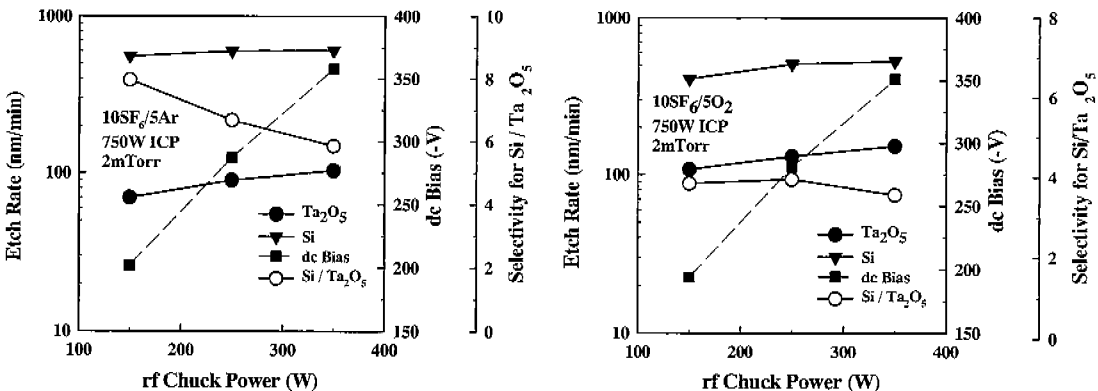


Fig. 3. Etch rates and selectivities for Si over Ta<sub>2</sub>O<sub>5</sub> in 10SF<sub>6</sub>/5Ar (left) and 10SF<sub>6</sub>/5O<sub>2</sub> (right) ICP discharges (750 W source power, 2 mTorr), as a function of rf chuck power.

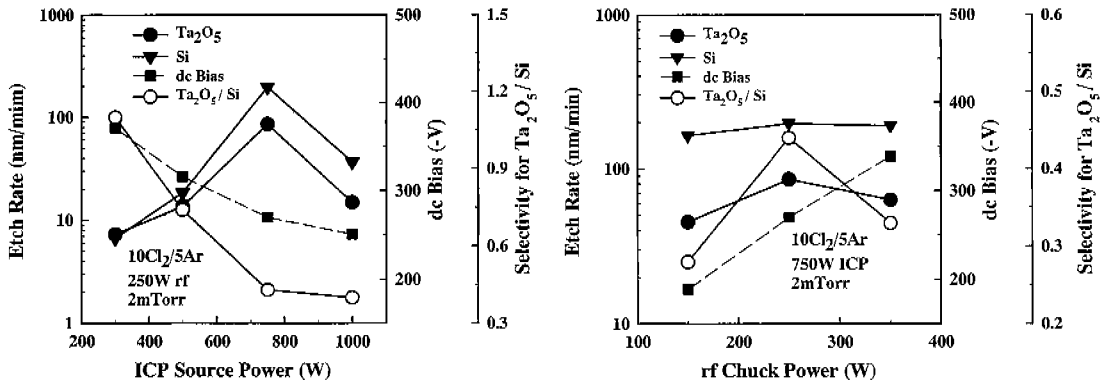


Fig. 5. Etch rates and selectivities for  $Ta_2O_5$  over Si in  $10Cl_2/5Ar$  ICP discharges as a function of source power (left) or rf chuck power (right).

assisted desorption of the reactive neutrals before they can react with the substrate [14]. The  $Ta_2O_5/Si$  selectivities again fall in the range of 0.3-1 over the whole range of conditions investigated.

### 3.2. The UV light enhancement of $Ta_2O_5$ dry etch rates

Figure 6 shows the  $Ta_2O_5$  etch rates in  $10SF_6/5Ar$  discharges as a function of ICP source power (left) and rf chuck power (right). At the low chuck power condition (100 W) the etch rate enhancement with UV illumination increases with source power and reaches a factor of approximately 2 in the range of 500~750 W. By sharp contrast, at higher rf chuck power conditions (Fig. 6, right) there is essentially no increase in etch rate as a result of UV illumination. At higher ion energies the etch rates decrease both with and without UV

illumination due to the ion-assisted removal of the reactive neutrals [14]. Several groups have reported that UV irradiation during dry etching dramatically enhances the etch rate of Cu in  $Cl_2$ -based high density plasmas, through transforming involatile  $CuCl_x$  etch products into more volatile species (e.g.  $Cu_2Cl_3$ ) and subsequent non-thermal desorption of these species. [15] In the case of  $Ta_2O_5$ , an analogous situation would involve photodesorption of  $TaF_x$  species since the oxygen should be removed as  $O_2$  or oxyfluorides.

In the case of  $Cl_2/Ar$  discharges, Fig. 7 shows that UV illumination did lead to faster  $Ta_2O_5$  etch rates at moderate source power (left) and rf chuck powers (right). This seems plausible from the following scenarios- at very high ion fluxes or energies, the  $TaCl_x$  etch product is being efficiently removed by sputter-assisted desorption whether or not the UV illumina-

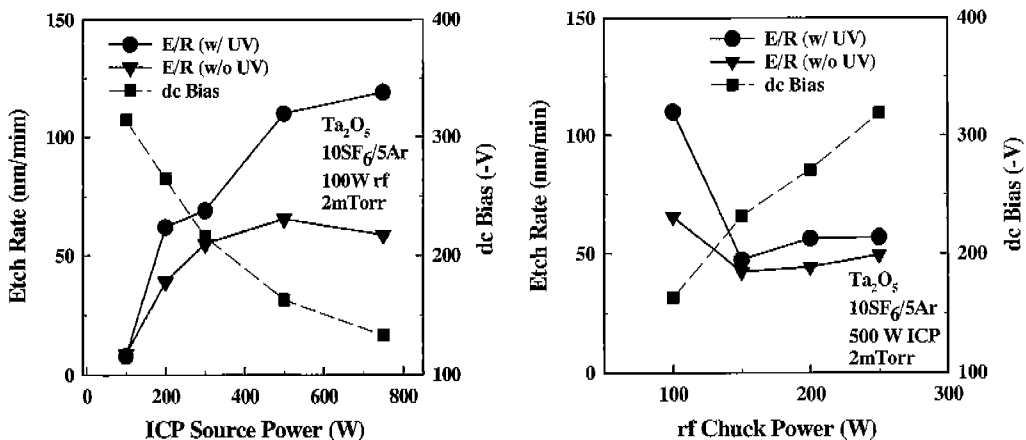


Fig. 6.  $Ta_2O_5$  etch rates with and without UV illumination in  $10SF_6/5Ar$  ICP discharges as a function of source power (left) or rf chuck power (right).

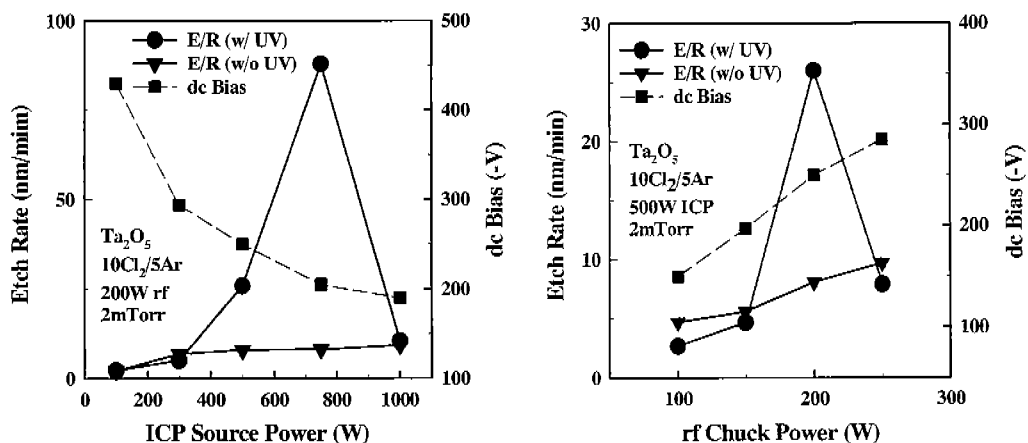


Fig. 7. Ta<sub>2</sub>O<sub>5</sub> etch rates with and without UV illumination in 10Cl<sub>2</sub>/5Ar ICP discharges as a function of source power (left) or rf chuck power (right).

tion is being used, whereas at very low ion fluxes or energies the etch rate may be controlled by diffusion of the etch products through a chlorinated selvedge layer. For the Cl<sub>2</sub>-based plasma chemistry the physical process dominates the etch process and UV illumination is less important over a broad range of conditions than with SF<sub>6</sub> plasma chemistry.

### 3.3. ICP etching of (Ba,Sr)TiO<sub>3</sub>

Figure 8 shows the rf chuck power dependence of (Ba,Sr)TiO<sub>3</sub> and LaNiO<sub>3</sub> etch rates in 10Cl<sub>2</sub>/5Ar (left) or 5CH<sub>4</sub>/10H<sub>2</sub>/5Ar (right) discharges at fixed source power (750 W) and pressure (2 mTorr). In 10Cl<sub>2</sub>/5Ar discharges, the etch rate of (Ba,Sr)TiO<sub>3</sub> increases with

the higher ion bombardment energy up to approximately 250 W rf chuck power and decreases thereafter. This trend is again, as mentioned earlier, ascribed to ion-assisted desorption of the adsorbed chlorine neutral before the etch products are formed on the surface. In the case of LaNiO<sub>3</sub> we do not observe the decrease in etch rate at higher rf chuck powers, suggesting the amount or stability of the adsorbed chlorine is different than for (Ba,Sr)TiO<sub>3</sub>. By sharp contrast, extremely low etch rates ( $\leq 10$  nm) were obtained for both (Ba,Sr)TiO<sub>3</sub> and LaNiO<sub>3</sub> in 5CH<sub>4</sub>/10H<sub>2</sub>/5Ar discharges and the etching is dominated by the physical sputtering since there is no apparent chemical contribution to the etching, with the results similar to those obtained with pure Ar plasmas.

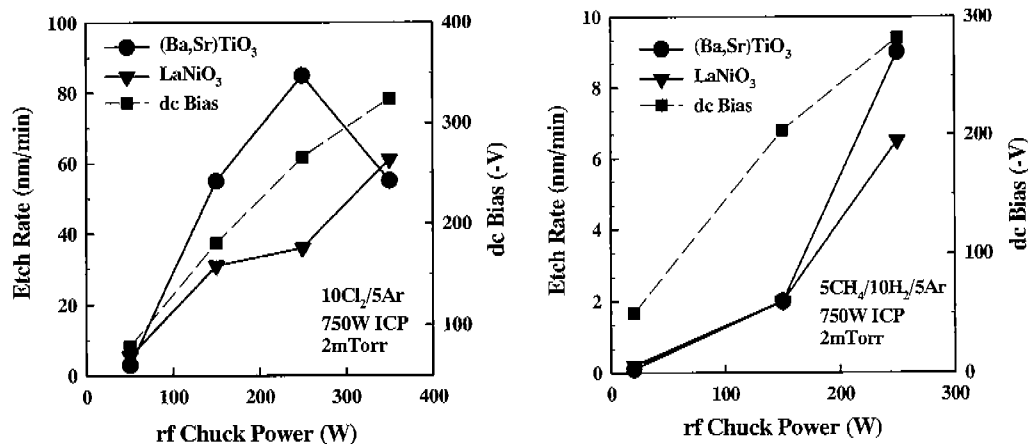


Fig. 8. Etch rates for (Ba,Sr)TiO<sub>3</sub> and LaNiO<sub>3</sub> in 10Cl<sub>2</sub>/5Ar (left) or 5CH<sub>4</sub>/10H<sub>2</sub>/5Ar (right) ICP discharges (750 W source power, 2 mTorr) as a function of rf chuck power.

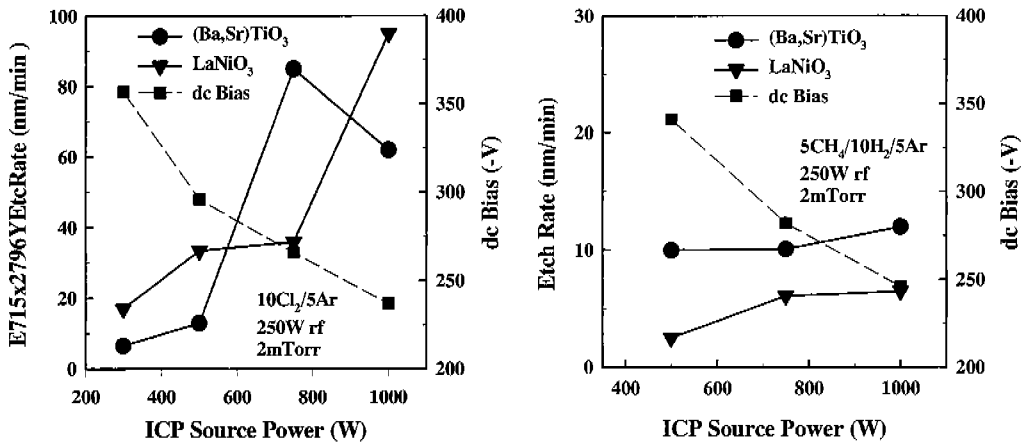


Fig. 9. Etch rates for (Ba,Sr)TiO<sub>3</sub> and LaNiO<sub>3</sub> in 10Cl<sub>2</sub>/5Ar (left) or 5CH<sub>4</sub>/10H<sub>2</sub>/5Ar (right) ICP discharges (250 W rf chuck power, 2 mTorr) as a function of source power.

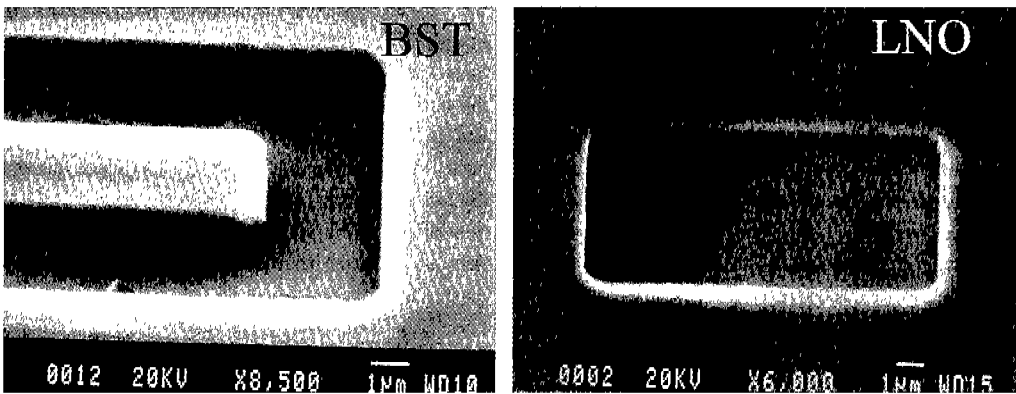


Fig. 10. SEM micrographs of features etched into (Ba,Sr)TiO<sub>3</sub> (left) and LaNiO<sub>3</sub> (right) using 10Cl<sub>2</sub>/5Ar ICP discharges (750 W source power, 250 W rf chuck power, 5 mTorr).

The influence of ICP source power on the etch rates of both (Ba,Sr)TiO<sub>3</sub> and LaNiO<sub>3</sub> in 10Cl<sub>2</sub>/Ar (left) or 5CH<sub>4</sub>/10H<sub>2</sub>/5Ar (right) discharges at fixed rf chuck power (250 W) and pressure (2 mTorr) is shown in Fig. 9. In high density plasma etching, increasing the source power suppresses the dc self-bias because of the higher conductivity of the plasma and this leads to two competing effects, namely an increase in ion flux but a decrease in ion energy. This competition is reflected in an initial increase in (Ba,Sr)TiO<sub>3</sub> etch rate in 10Cl<sub>2</sub>/5Ar mixtures, followed by a decrease when the dc self-bias falls below approximately 270 V. Once again the LaNiO<sub>3</sub> etch rate shows different behavior, with a continuing increase in etch rate over the range of source powers we investigated. 5CH<sub>4</sub>/10H<sub>2</sub>/5Ar mixtures again produced low etch rates for both materials under all conditions most likely due to the deposition

of a polymer layer from the CH<sub>4</sub>, which may act to shield the surface from the ion bombardments.

Since high density plasma etching for (Ba,Sr)TiO<sub>3</sub> and LaNiO<sub>3</sub> in 10Cl<sub>2</sub>/5Ar discharges is ion-driven etching under all conditions investigated, highly anisotropic features can be formed with a minimized mask erosion, and SEM micrographs of the features etched into (Ba,Sr)TiO<sub>3</sub> (left) and LaNiO<sub>3</sub> (right) are shown in Fig. 10. A 7 μm thick photoresist mask was used as a mask material and about one-third of the resist remained at the completion of the etching.

#### 4. Conclusions

A number of different inductively coupled plasma chemistries (SF<sub>6</sub>/O<sub>2</sub>, SF<sub>6</sub>/Ar and Cl<sub>2</sub>/Ar) were examined

for dry etching of Ta<sub>2</sub>O<sub>5</sub> and maximum etch rates ~120 nm/min were achieved in fluorine- or chlorine-based mixtures. The etch selectivities for Si over Ta<sub>2</sub>O<sub>5</sub> of ~6 were achieved in SF<sub>6</sub> ICP discharges since Ta<sub>2</sub>O<sub>5</sub> etch rates were always slower than those of Si and the reverse selectivity was not achieved.

The UV illumination during ICP etching of Ta<sub>2</sub>O<sub>5</sub> in both SF<sub>6</sub>/Ar and Cl<sub>2</sub>/Ar plasma chemistries produces significant enhancements in etch rates. The increased etch rates are likely due to photoassisted desorption of the TaF<sub>x</sub> and TaCl<sub>x</sub> etch products. The use of UV illumination is an alternative to employing elevated temperatures during etching to increase the volatility of the etch products and may find application where the thermal budget should be minimized during processing.

Two common semiconductor plasma chemistries, namely Cl<sub>2</sub>/Ar and CH<sub>4</sub>/H<sub>2</sub>/Ar, have been examined for dry etching of thin films of (Ba,Sr)TiO<sub>3</sub> and LaNiO<sub>3</sub>. The etching in both chemistries is physically dominated, but only Cl<sub>2</sub>/Ar produces reasonable etch rates, and smooth and anisotropic pattern transfer was performed.

## References

- [1] B.C. Lai and J.Y. Lee, Leakage Current Mechanism of Metal-Ta<sub>2</sub>O<sub>5</sub>-metal Capacitors for Memory Device Applications, *J. Electrochem. Soc.* 146 (1999) 266.
- [2] K.W. Kwon, C.S. Kang, S.O. Park, H.K. Kang and S.T. Ahn, Thermally Robust Ta<sub>2</sub>O<sub>5</sub> Capacitor for the 256-Mbit DRAM, *IEEE Trans. Electron. Dev.* 43 (1996) 919.
- [3] S.O. Kim and H.J. Kim, Fabrication of *n*-metal-oxide Semiconductor Field Effect Transistor with Ta<sub>2</sub>O<sub>5</sub> Gate Oxide Prepared by Plasma Enhanced Metalorganic Chemical Vapor Deposition, *J. Vac. Sci. Technol. B* 12 (1994) 3006.
- [4] H. Shimada and T. Ohmi, Current Drive Enhancement by Using High-permittivity Gate Insulator in SOI MOSFET's and its Limitation, *IEEE Trans. Electron. Dev.* 43 (1996) 431.
- [5] K. Kukli, J. Ihances, M. Ritala and M. Leskela, Properties of Ta<sub>2</sub>O<sub>5</sub>-based Dielectric Nanolaminates Deposited by Atomic Layer Epitaxy, *J. Electrochem. Soc.* 144 (1997) 300.
- [6] A.K. Chu, Y.S. Huang and S.H. Tang, Room-temperature Radio Frequency Sputtered Ta<sub>2</sub>O<sub>5</sub> : A New Etch Mask for Bulk Silicon Dissolved Processes, *J. Vac. Sci. Technol. B* 17 (1999) 455.
- [7] K. Kukli, J. Ihances, M. Ritala and M. Leskela, Tailoring the Dielectric Properties of HfO<sub>2</sub>-Ta<sub>2</sub>O<sub>5</sub> nanolaminates, *Appl. Phys. Lett.* 68 (1996) 3737.
- [8] Y. Kuo, Reactive Ion Etching of Sputter Deposited Tantalum Oxide and its Etch Selectivity to Tantalum, *J. Electrochem. Soc.* 139 (1992) 579.
- [9] S. Seki, T. Unagami, and B. Tsujiyama, Reactive Ion Etching of Tantalum Pentoxide, *J. Electrochem. Soc.* 130 (1983) 2505.
- [10] S.M. Hong, S.M. Rhim, H.J. Bak and O.K. Kim, Deposition Characteristics of (Ba,Sr)RuO<sub>3</sub> Thin Films Prepared by Ultrasonic Spraying Deposition, *J. Kor. Assoc. Crystal Growth* 11 (2001) 111.
- [11] C.R. Cho, D.A. Payne and S.L. Cho, Solution Deposition and Heteroepitaxial Crystallization of LaNiO<sub>3</sub> Electrodes for Integrated Ferroelectric Devices, *Appl. Phys. Lett.* 71 (1997) 3013.
- [12] M.S. Chen, T.B. Wu and J.-M. Wu, Effect of Textured LaNiO<sub>3</sub> Electrode on the Fatigue Improvement of Pb(Zr<sub>0.53</sub>Ti<sub>0.47</sub>)O<sub>3</sub> Thin Films, *Appl. Phys. Lett.* 68 (1996) 1430.
- [13] A. Li, C. Be, P. Lu, D. Wu, S. Xing and N. Ming, Fabrication and Electrical Properties of Sol-gel Derived BaTiO<sub>3</sub> Films with Metallic LaNiO<sub>3</sub> Electrode, *Appl. Phys. Lett.* 70 (1997) 1616.
- [14] R.J. Shul, G.B. McClellan, R.D. Briggs, D. Riegler, S.J. Pearton, C.R. Abernathy, J.W. Lee, C. Constantine and C. Barrat, High-density Plasma Etching of Compound Semiconductors, *J. Vac. Sci. Technol. A* 15 (1997) 633.
- [15] K.-S. Choi and C.-H. Han, Low Temperature Copper Etching Using an Inductively Coupled Plasma with Ultraviolet Light Irradiation, *J. Electrochem. Soc.* 145 (1998) L37.