Effect of Calcination Temperature on the Structure and Electrochemical Performance of LiMn_{1.5}Ni_{0.5}O₄ Cathode Materials

Seo Hee Ju and Dong-Won Kim*

Department of Chemical Engineering, Hanyang University, Seoul 133-791, Korea. *E-mail: dongwonkim@hanyang.ac.kr Received September 3, 2012, Accepted October 2, 2012

Spinel LiMn_{1.5}Ni_{0.5}O₄ cathode powders with different morphologies were synthesized by a co-precipitation method using oxalic acid. The calcination temperature affected the morphologies, crystalline structure and electrochemical properties of the LiMn_{1.5}Ni_{0.5}O₄ powders. The LiMn_{1.5}Ni_{0.5}O₄ powders obtained at a calcination temperature of 850 °C exhibited the highest initial discharge capacity with good capacity retention and high rate capability.

Key Words : Cathode, Lithium nickel manganese oxide, Lithium-ion battery, Calcination temperature, Electrochemical performance

Introduction

Lithium-ion batteries are attractive power sources for portable electronic devices, electric vehicles and energy storage systems due to their high energy density and long cycle life.¹ Lithium manganese oxide (LiMn₂O₄) with a spinel structure has been considered one of the promising cathode materials for rechargeable lithium batteries, owing to its low cost, environmental friendliness, high voltage operation and high safety. However, stoichiometric LiMn₂O₄ materials exhibit large capacity fade on cycling because of the Jahn-Tell distortion of trivalent Mn ions and the dissolution of Mn into the liquid electrolyte, especially at elevated temperature.²⁻⁶ In order to overcome these problems, several cationic substitutions to produce LiMn_{2-y}M_yO₄ (M =Cr, Fe, Co, Ni, and Cu) have been studied in the literature.⁷⁻¹³ Among these materials, LiMn_{1.5}Ni_{0.5}O₄ is of great interest as a cathode material for 5 V lithium-ion cells because of its high specific capacity and its dominant plateau at around 4.7 V, which arise from the presence of a Ni²⁺/Ni⁴⁺ redox pair.¹⁴⁻¹⁷ Various synthetic methods such as sol-gel, molten salt, co-precipitation and composition emulsion drying have been applied to the preparation of cathode active materials for lithium-ion batteries. The co-precipitation method is known as a simple way to prepare fine, highly pure and homogeneous powders of single or multi-component oxide materials.¹⁸⁻²⁰

In this study, we synthesize spinel $LiMn_{1.5}Ni_{0.5}O_4$ cathode powders using the co-precipitation method at different calcination condition in order to achieve the best electrochemical properties. We investigate the effect of the calcination temperature on the morphologies, crystalline structure and electrochemical properties of spinel $LiMn_{1.5}Ni_{0.5}O_4$ cathode powders.

Experimental

 $LiMn_{1.5}Ni_{0.5}O_4$ cathode powders were synthesized by coprecipitation using oxalic acid. A stoichiometric amount of lithium acetate dihydrate [Li(CH₃COO)·2H₂O, Aldrich], manganese(II) acetate tetrahydrate [Mn(CH₃COO)₂·4H₂O, Aldrich] and nickel(II) acetate tetrahydrate [Ni(CH₃COO)₂·4H₂O, Aldrich] were dissolved in de-ionized water. The solution was stirred continuously under heating at 60 °C. The oxalic acid solution was added to the homogeneous solution drop by drop. The mole ratio of oxalic acid to metal was 5.0. The solution was stirred at 90 °C for 2 h and further dried overnight at 110 °C. The dried precipitate was preheated at 450 °C for 3 h to decompose the organic components. The precursor powders were calcined at different temperatures (700, 800, 850, 900, 950, 1000 °C) for 15 h in air atmosphere. The morphological characteristics of the LiMn_{1.5}Ni_{0.5}O₄ powders were investigated using scanning electron microscopy (SEM, JEOL JSM 6701). Energy dispersive spectroscopy (EDS) and elemental mapping were performed using the same instrument. Powder X-ray diffraction (XRD, Philips X'PERT MPD) using Cu Ka radiation was used to identify the crystalline phase of the LiMn_{1.5}Ni_{0.5}O₄ powders. The cathode was prepared by coating an N-methyl pyrrolidone (NMP)-based slurry containing LiMn_{1.5}Ni_{0.5}O₄ powder, poly-(vinylidene fluoride) (PVdF) and super-P carbon (85:7.5:7.5 by weight) on an aluminum foil. The thicknesses of the electrodes ranged from 50 to 55 µm after roll pressing, and their active mass loading corresponded to a capacity of about 1.2 mAh cm⁻². The lithium electrode consisted of a 150 µmthick lithium foil that was pressed onto a copper current collector. A CR2032-type coin cell composed of lithium anode, a polypropylene separator (Celgard 2400) and LiMn_{1.5}Ni_{0.5}O₄ cathode was assembled with an electrolyte solution. The liquid electrolyte used in this study was 1 M $LiPF_6$ in ethylene carbonate (EC)/dimethyl carbonate (DMC) (1:1 by volume, battery grade, Soulbrain Co. Ltd.). All cells were assembled in a dry box filled with argon gas. Charge and discharge cycling tests of the Li/LiMn_{1.5}Ni_{0.5}O₄ cells were conducted at different current density over a voltage range of 3.0-4.9 V with battery testing equipment at room temperature.

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Results and Discussion

The LiMn_{1.5}Ni_{0.5}O₄ powders with different morphologies were prepared by co-precipitation method with oxalic acid at different calcination temperatures. The effect of the calcination temperature on the morphologies of the LiMn₁ $_{5}Ni_{0} _{5}O_{4}$ powders is shown in Figure 1. The LiMn_{1.5}Ni_{0.5}O₄ powders calcined at 700 °C exhibited nano-rod morphologies and aggregation of the powders, whereas powders calcined at 800 °C exhibited a well-faced crystallized spinel phase instead of a rod shape. As the calcination temperature increased, the particle size increased and the particle shape and distribution became more uniform. However, the particles calcined at 1000 °C had a larger size of several microns with aggregated morphologies between the particles. As a result, the LiMn_{1.5}Ni_{0.5}O₄ powders calcined at temperature ranges of 800 to 900 °C are well-distributed crystalloids with little agglomeration.

Figure 2 shows the XRD patterns of the LiMn_{1.5}Ni_{0.5}O₄ cathode powders calcined at different temperatures. The crystalline phase of all the powders can be identified to be a spinel structure indexed by cubic *Fd3m* in which lithium ions occupy the tetragonal (8a) sites, transition metals (Mn and Ni) are located at the octahedral (16d) sites, and oxygen atoms reside in 32e sites.²¹ This indicates that the Mn site in the LiMn₂O₄ is partially substituted by Ni. Scherrer's formula given in Eq. (1) was used to determine the mean crystallite size (*t*) of the powder:

$$t = (0.94 \lambda) / (B \cos \theta_B) \tag{1}$$

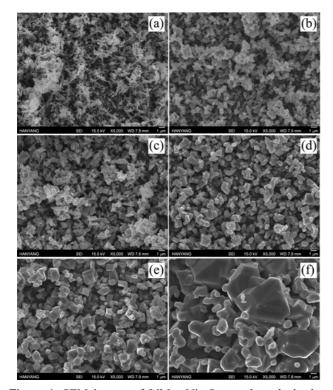


Figure 1. SEM images of $LiMn_{1.5}Ni_{0.5}O_4$ powders obtained at different calcination temperatures. (a) 700, (b) 800, (c) 850, (d) 900, (e) 950 and (f) 1000 °C.

531 1000 °C 950 °C ntensity (a. u.) 900 °C 850 °C 800 °C 700 °C 20 30 40 50 60 70 10 80 20

Figure 2. XRD patterns of $LiMn_{1.5}Ni_{0.5}O_4$ powders prepared at different calcination temperatures.

where λ is the wavelength of the X-ray used, θ_B is the Bragg angle of the diffraction peak considered, and B is the width at an intensity equal to half of the maximum intensity.²² When the calcination temperature was increased from 700 to 1000 °C, the mean crystallite size of the cathode powders increased from 28 to 35 nm. By increasing the calcination temperature, the peak widths of the LiMn_{1.5}Ni_{0.5}O₄ powders become narrower and the relative intensities of the characteristic peaks gradually increase, indicating that a better crystalline structure of LiMn1.5Ni0.5O4 was obtained. However, some very tiny peaks are observed around 37.5, 43.7 and 63.6°, which correspond to the weak impurity phases of Li_xNi_{1-x}O.^{23,24} According to the previous reports,²⁵ LiMn_{1.5}Ni_{0.5}O₄ can loss oxygen and disproportionate to a mixture of a spinel phase and a secondary phase LixNi1-xO because of partial reduction of Mn⁴⁺ to Mn³⁺ at elevated temperature. Thus, the formation of the Li_xNi_{1-x}O phase may result in a decrease of the Ni content and the Mn valence in the spinel phase. This result indicates that heat treatment at higher temperature results in more oxygen deficiency,

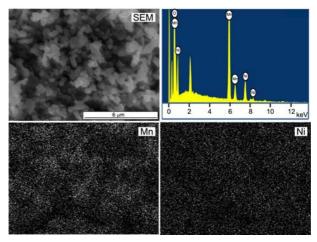


Figure 3. EDS spectrum of $LiMn_{1.5}Ni_{0.5}O_4$ powders calcined at 850 °C, and dot mappings for the Mn and Ni atoms in the corresponding $LiMn_{1.5}Ni_{0.5}O_4$ powders.

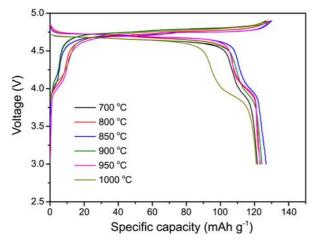


Figure 4. Initial charge and discharge curves of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different calcination temperatures (0.5C rate, cut-off voltage: 3.0-4.9 V).

formation of Mn³⁺ and the occurrence of a secondary phase.

Figure 3 shows the EDS spectrum and dot mapping images of the $LiMn_{1.5}Ni_{0.5}O_4$ cathode powders calcined at 850 °C. In the dot mapping, the Mn and Ni components were observed to be well-dispersed inside the submicron-sized $LiMn_{1.5}Ni_{0.5}O_4$ powders. From the EDS spectrum shown in figure, the atomic ratio of manganese and nickel was calculated to be about 3.01:1, which is consistent with the chemical composition of the LiMn_{1.5}Ni_{0.5}O_4 powders.

Figure 4 shows the initial charge and discharge curves of LiMn_{1.5}Ni_{0.5}O₄ cathode materials calcined at different temperatures. The Li/LiMn_{1.5}Ni_{0.5}O₄ cells were subjected to charge and discharge cycling in the voltage range of 3.0-4.9 V at a constant current rate of 0.5C. The initial discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ active materials ranged from 122 to 126 mAh g^{-1} , depending on the calcination temperature. The highest discharge capacity (126 mAh g⁻¹) was delivered by LiMn_{1.5}Ni_{0.5}O₄ powders calcined at 850 °C. Discharge capacities vs cycle number of the LiMn_{1.5}Ni_{0.5}O₄ cathode materials between 3.0 and 4.9 V are shown in Figure 5. LiMn_{1.5}Ni_{0.5}O₄ cathode material obtained at 850 °C exhibited high discharge capacity and good capacity retention (93% of initial discharge capacity after 100 cycles). As explained earlier, high calcination temperature may result in oxygen loss, Ni deficiency of LiMn_{1.5}Ni_{0.5}O₄ and formation of LixNi1-xO impurities,24 which decrease the discharge capacity and deteriorate the cycling stability of the spinel LiMn_{1.5}Ni_{0.5}O₄ materials calcined at temperature higher than 850 °C. On the other hand, the integrated crystalline phase for structural stability is not well formed when the calcination temperature is below 850 °C. As a result, the discharge capacities are low in the LiMn_{1.5}Ni_{0.5}O₄ materials calcined at temperature lower than 850 °C. It is well known that the dissolution of transition metals suffering from unavoidable HF in electrolyte and the formation of LiF on the surface of cathode materials due to the electrolyte decomposition are mainly responsible for the capacity loss of electrode materials during cycling.²⁵⁻²⁸ When the calcination temperature is

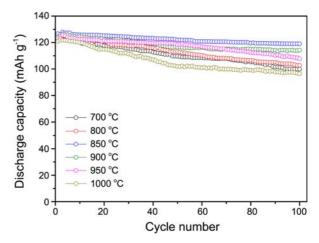


Figure 5. Discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials as a function of cycle number (0.5C rate, cut-off voltage: 3.0-4.9 V).

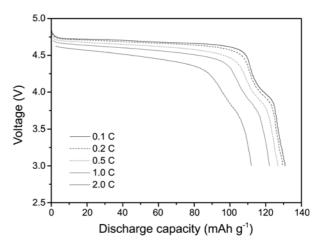


Figure 6. Discharge curves of the LiMn_{1.5}Ni_{0.5}O₄ cathode materials prepared at 850 °C, as a function of the C rate. The charging current rate was 0.1C with a 4.9 V cut-off.

below 850 °C, the LiMn_{1.5}Ni_{0.5}O₄ cathode materials have smaller particle size and larger specific surface area, thus higher dissolution of transition metals and electrolyte oxidation rate on its surface can be expected during cycling. Accordingly, the capacity retention behaviors of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at 700 and 800 °C are not as good as that of LiMn_{1.5}Ni_{0.5}O₄ obtained at 850 °C. As a result, the LiMn_{1.5}Ni_{0.5}O₄ cathode materials calcined at 850 °C exhibited the best electrochemical performance in consideration of discharge capacity and capacity retention.

We evaluated the rate capability of the LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different temperatures. The Li/LiMn_{1.5}Ni_{0.5}O₄ cells were charged to 4.9 V at a constant current rate of 0.1C and discharged at different current rates ranging from 0.1 to 2.0C. The discharge curves of the LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at 850 °C are shown in Figure 6. Both voltage and discharge capacity decreased gradually as current rate increased. At 2.0C rate, the cell delivered a relatively high discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different discharge capacities of LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained at different

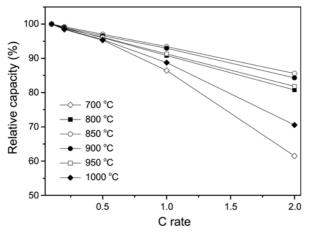


Figure 7. Relative capacities of $LiMn_{1.5}Ni_{0.5}O_4$ cathode materials prepared at different calcination temperatures, as a function of the C rate.

calcination temperatures, as a function of the C rate. The relative capacity is defined as the ratio of the discharge capacity at a specific C rate to the discharge capacity delivered at 0.1C rate. This result shows that the calcination temperature also affects the high rate performance. At a discharge rate of 2C, the LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained from 700, 800, 850, 900, 950 and 1000 °C delivered relative capacities of 61, 81, 86, 84, 82 and 71%, respectively. As for LiMn_{1.5}Ni_{0.5}O₄ cathode materials obtained below 850 °C, their lower crystallinity may be detrimental to their high rate performance. Moreover, smaller particle size with large specific surface area may give more chances to side reactions such as Mn³⁺ dissolution in the electrolyte. On the other hand, the LiMn_{1.5}Ni_{0.5}O₄ cathode materials calcined at high temperature have longer diffusion path for the intercalation and deintercalation of lithium ions, which deteriorate their cycling performance at high current rates. Hence, the LiMn_{1.5}Ni_{0.5}O₄ cathode materials calcined at 850 °C exhibited the best high rate performance, which can be ascribed to their fine size and well-defined crystalline structure.

Conclusion

LiMn_{1.5}Ni_{0.5}O₄ cathode powders were prepared at different calcination temperatures using the co-precipitation method. We investigated the effect of the calcination temperature on the morphologies, crystalline structure and electrochemical properties of the LiMn_{1.5}Ni_{0.5}O₄ powders. The LiMn_{1.5}Ni_{0.5}O₄ powders obtained from 800 to 900 °C showed a well-crystallized spinel phase and uniform shape with fine size. The LiMn_{1.5}Ni_{0.5}O₄ cathode materials calcined at 850 °C delivered the highest discharge capacities and exhibited good cycling characteristics in terms of the capacity retention and rate capability.

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